

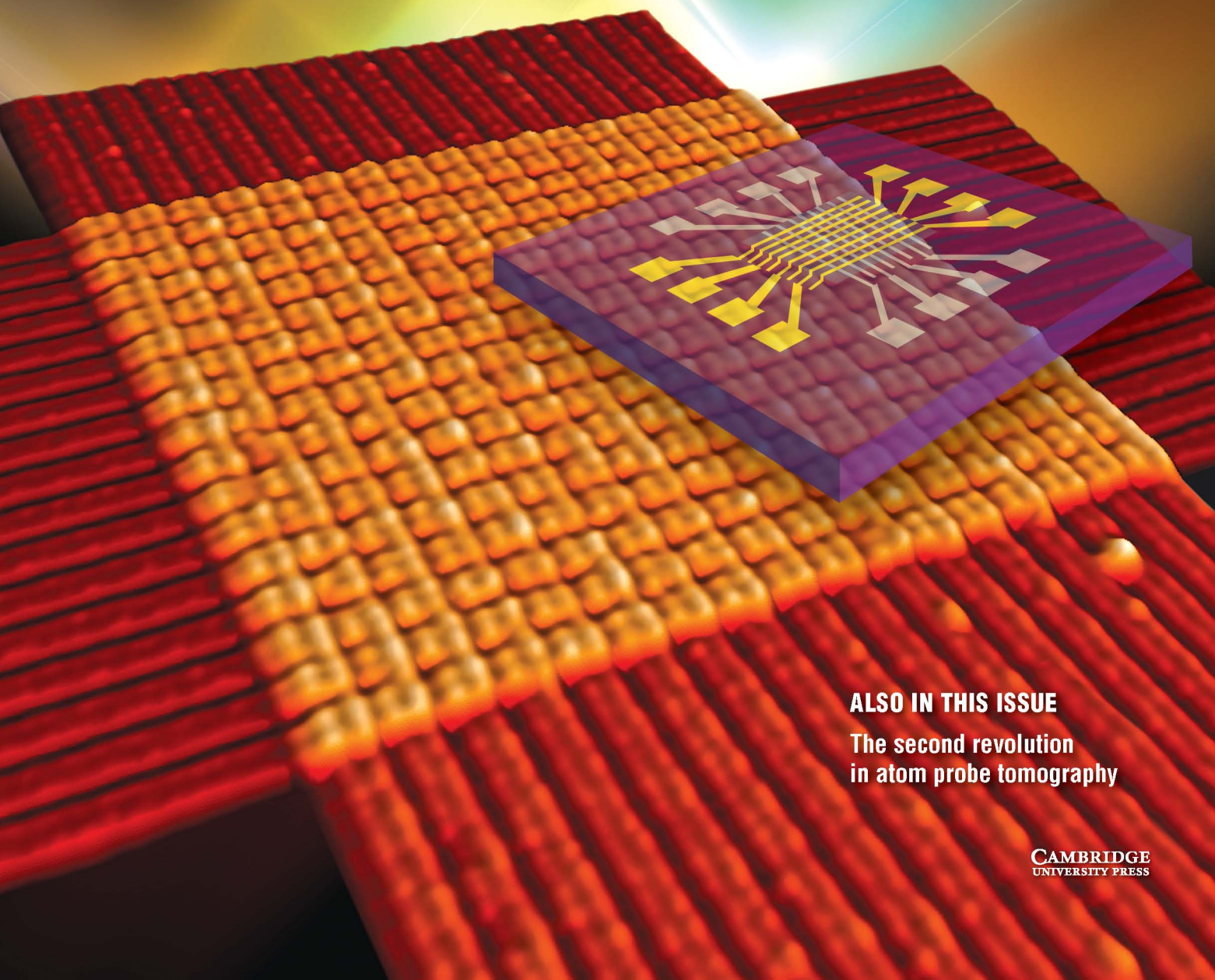
February 2012 Vol. 37 No. 2
www.mrs.org/bulletin

MRS Bulletin



MATERIALS RESEARCH SOCIETY
Advancing materials. Improving the quality of life.

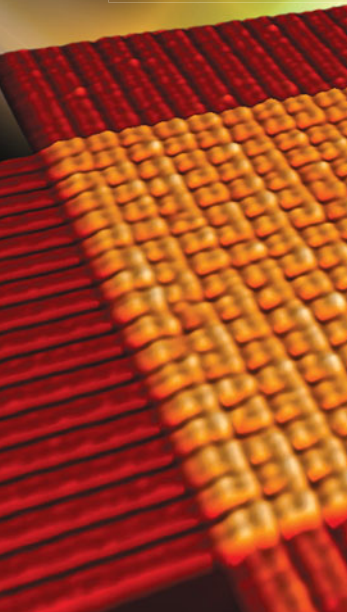
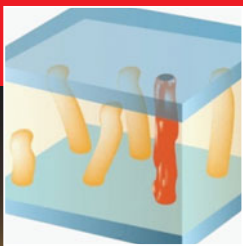
Resistive switching phenomena in thin films: Materials, devices, and applications



ALSO IN THIS ISSUE

The second revolution
in atom probe tomography

CAMBRIDGE
UNIVERSITY PRESS



Organic resistive nonvolatile memory materials

Takhee Lee and Yong Chen

Resistive memory devices based on organic materials that can be configured to two or more stable resistance states have been extensively explored as information storage media due to their advantages, which include simple device structures, low fabrication costs, and flexibility. Various organic-based materials such as small molecules, polymers, and composite materials have been observed to show bistability. This review provides a general summary about the materials, structures, characteristics, and mechanisms of organic resistive memory devices. Several critical strategies for device fabrication, performance enhancement, and integrated circuit architectures are also discussed.

Introduction

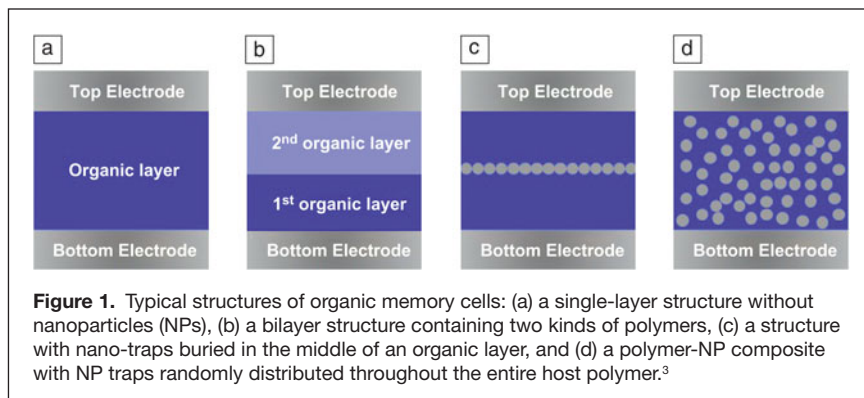
A memory device composed of organic materials sandwiched between two electrodes possesses at least two stable resistance states controlled by an external electrical stimulus.^{1–3} Organic materials provide many advantages, including simple device structures, low fabrication costs, printability, and flexibility. The device resistance states can be read nondestructively, and no electrical power is required to maintain the states, indicating a nonvolatile memory.^{1–3} Furthermore, essential requirements for high performance memory, such as a high ON/OFF ratio, long endurance and retention characteristics, and a fast switching speed, have been experimentally verified.^{4,5} Although it is difficult to identify the device operating mechanisms, in-depth studies of charge conduction mechanisms have contributed to the device physics that underlies the switching phenomena. Structural and electrical optimizations^{5,6} have also been applied successfully to improve device performance. The main focus of this review is to provide a general summary about the materials, structures, characteristics, and mechanisms of organic resistive memory devices. We also introduce several critical strategies for device integration and advanced circuit architectures.

Materials, structures, and characteristics of organic switching devices

An organic resistive memory device consists of organic layers sandwiched between bottom and top electrodes. Various organic materials have shown switching resistances in response to

applied voltages.^{1,7–36} Organic materials that exhibit conductance switching include small molecules,^{17–20,36} polymers,^{1,21–27} and composites containing nanoparticles (NPs).^{8–10,28–35} Specifically, the conductance switching of single and bundled phenylene ethynylene oligomers isolated in matrices of alkanethiolate monolayers was caused by conformational changes in the molecules or bundles.¹⁷ A reliable nonvolatile memory device using a polyfluorene-derivative single-layer film has been characterized.²⁴ Nonvolatile resistive memory effects of gold nanoparticles embedded in the conducting polymer poly(4-n-hexylphenyldiphenylamine) also has been investigated using admittance spectroscopy.³¹ In particular, the following four device structures have been implemented (**Figure 1**): (1) a single-layer structure containing only one type of organic material, (2) a bilayer structure containing two types of organic materials, (3) a trilayer structure in which nano-traps for charge carriers are sandwiched between two organic layers, and (4) spin-cast polymer-NP blends in which nano-traps are randomly distributed throughout the entire region of the host matrix. Among these device structures, the single-polymer-layer structure is more attractive due to the simplicity of device fabrication. Organic-based composite materials are commonly used to induce resistive switching.^{8–10,28–35,37} Polymer-blend systems containing organic or inorganic NPs mainly constitute the active materials of resistive memory devices. Resistance switching in these materials is caused by the introduction of NPs into the polymer matrix.^{8,9,29–31,33,35,37} The concentration and distribution

Takhee Lee, Department of Physics and Astronomy, Seoul National University, Korea; tlee@snu.ac.kr
Yong Chen, University of California, Los Angeles, USA; yongchen@seas.ucla.edu
DOI: 10.1557/mrs.2012.4



of NPs in a blend system are important fabrication factors and must be carefully controlled during solution preparation.^{8,11,28,37} The active matrix architecture^{38–41} (i.e., crossbar organic resistive memory combined with a diode or transistor) is also essential for practical memory applications.

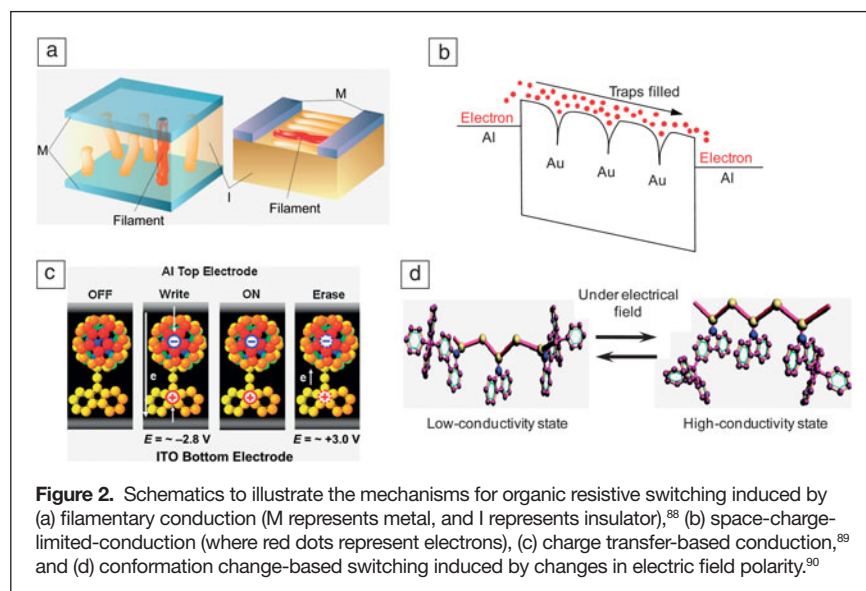
Mechanisms of resistive switching

The conduction mechanism in organic materials is complex because the amorphous structure of the organic materials cannot be explained adequately on the basis of ordered inorganic materials.⁴² The conduction in organic materials is normally explained by intrinsic charge carrier generation in the organic materials and charge carrier injection from electrodes to organic materials. Intrinsic carrier density in organic materials is usually very low, and carriers are usually trapped by localized states, which makes it difficult to understand charge transport behavior. A number of conduction mechanisms such as Ohmic conduction, Schottky emission, thermionic emission, space charge-limited current (SCLC), tunneling current, ionic conduction, hopping conduction, and impurity conduction have been utilized to explain the conduction process in organic materials.³

When the current in the ON state is highly localized to a small fraction of the device area, the conduction phenomenon is called filamentary conduction (**Figure 2a**).^{43,44} Two kinds of filamentary conduction have been suggested: One is associated with the carbon-rich filaments formed by the local degradation of organic films.^{45,46} The other is related to the metallic bridges that result from the migration of electrodes.^{43,47} When filaments are formed in an organic device, the ON state shows temperature-independent electrical behavior or metallic I - V characteristics.²² It also exhibits a device area-independent behavior, which is due to the formation of much smaller filaments.^{22,46} Highly localized current images of ON state materials have also been used to support filamentary switching mechanisms.¹³ As an example of filamentary switching, Cu ions that penetrated into a poly(3-hexylthiophene) (P3HT) layer under

forward bias resulted in filament formation in a Cu/P3HT/Al device, which was experimentally observed by secondary ion mass spectroscopy analysis.⁴⁸ Similar metallic filaments were also formed in poly(ethylenedioxythiophene):poly(styrenesulfonate) (PEDOT:PSS) or poly(4-hydroxystyrene) composite materials that contain homogeneously dispersed Au clusters.⁴⁹ Nanoscale memory devices can be easily achieved in filamentary-based systems because switching is possible even in highly localized regions.

Another switching mechanism is related to SCLC. For instance, a trap-filled SCLC model was proposed to explain the resistive switching effect in Al/polystyrene (PS) + Au NPs/Al device (seen in **Figure 2b**). Therefore, Au NPs were treated as traps in the PS. When the electrode/organic material contact is Ohmic, the accumulation of carriers near the electrode builds up a space charge. Mutual repulsion between individual charges restricts the total charge injected into the organic material, resulting in SCLC.^{50,51} The electrical switching in some organic materials has been reported to be associated with space charges and SCLC.²⁵ For example, hysteresis characteristics in the I - V curves of a ITO/poly [3-(6-methoxyhexyl)thiophene] (P3OMe)/Al structure were observed.²⁵ Space charges accumulate at the metal/polymer interfaces and induce SCLC.⁵² The stored charges control charge injection and lead to hysteresis in the I - V curve. It is also observed that current abruptly increases at a threshold voltage, reaches the maximum value, and then passes through a negative differential region to the minimum value. It was proposed that the active mechanism is similar to that observed in inorganic metal-insulator-metal diodes by Simmons and Verderber.⁵³ Indeed, similar switching behaviors have been reported for many different materials in the same



trilayer structure,^{12,54,55} which provides insights into the switching mechanisms.

Switching may also be induced by charge transfer-based (Figure 2c) and conformation change-based conduction (Figure 2d). Charge transfer (CT) occurs in electron donor-acceptor systems in which electrical charges are partially transferred from the donor to the acceptor moiety.^{7,10,33,56} The electrical memory effect of CT complexes was observed in Cu-tetracyanoquinodimethane (TCNQ),⁵⁶ where Cu is the donor and TCNQ is the organic acceptor. Raman spectroscopy showed that the TCNQ anions in the pristine high resistance film were replaced by neutral TCNQ molecules in the low resistance state.⁵⁷ Chu et al.⁷ also observed that the bistable resistance states of a [6,6]-phenyl-C61-butyric acid methyl ester (PCBM) and tetrathiofulvalene (TTF) dispersed in a polystyrene matrix were due to a charge transfer effect between the TTF and PCBM in the film. A strong electric field facilitated the transfer of electrons from the highest occupied molecular orbital of TTF to the lowest unoccupied molecular orbital of PCBM, producing a positive charge in TTF and a negative charge in PCBM. A sharp increase in conductivity occurred after the charge transfer.⁷

Conductance switching and memory effects of molecular electronic devices often arise from electrically or optically induced conformational changes in molecules or molecular bundles.^{17,58} For example, the Rose Bengal molecule in supramolecular matrices of polyelectrolytes has shown conductance switching.³⁶ A similar effect was also proposed for the resistive switching of poly(2-(9H-carbazol-9-yl)ethyl methacrylate) (PCz) and poly(9-(2-((4-vinylbenzyl)oxy)ethyl)-9H-carbazole) (PVBCz), which are non-conjugated polymers containing electron-donating carbazole pendant groups.^{23,59} Upon application of an electric field, randomly oriented carbazole groups in the organic materials were rearranged into face-to-face conformations, thus facilitating carrier delocalization and transport and producing a high-conductance state. Ionic conduction-based resistive switching has been also reported.^{32,60} Perhaps, the exact switching mechanisms may depend on the organic materials used, that is, a unified switching model to describe the switching mechanism of various organic materials does not exist, and more than one switching mechanism might work simultaneously in an organic memory device.

Device performance and requirements

The performance of memory devices is specified by their storage capacity, ON/OFF ratio, speed to write or erase, cycling endurance, retention time, and fabrication cost. The value of the ON/OFF ratio determines whether or not the circuitry can

distinguish between high and low resistance values. Researchers usually emphasize the value of the ON/OFF ratio measured in single cells, but the statistical distribution of the ratios measured from many cells over many cycles should be considered to be more important. Narrow distribution of the resistances at each state and large gap between the states are considered to be necessary and challenging for organic memory circuits.

The series of plots shown in **Figure 3** demonstrate typical nonvolatile memory performance of organic memory devices fabricated using a single layer of polyfluorene-derivative (denoted as WPF-oxy-F, chemical structure shown in the inset of Figure 3a) film. Figure 3a shows I - V data measured from an organic memory device using a single layer of WPF-oxy-F film (left plot of Figure 3a),²⁴ which shows bipolar nonvolatile memory behavior with an ON/OFF ratio of $\sim 10^3$ at 0.3 V (right plot of Figure 3a). Figure 3b is a cumulative plot showing device uniformity in memory arrays.⁶¹ The inset in Figure 3b schematically shows an array of memory devices with an 8×8 crossbar structure, where red cells represent 60 functional switchable cells, and green cells represent 4 non-switchable cells. The distributions of both ON (low resistance state [LRS]) and OFF (high resistance state [HRS]) are within an order of magnitude, and the difference between HRS and LRS was found to be more than two orders of magnitude.

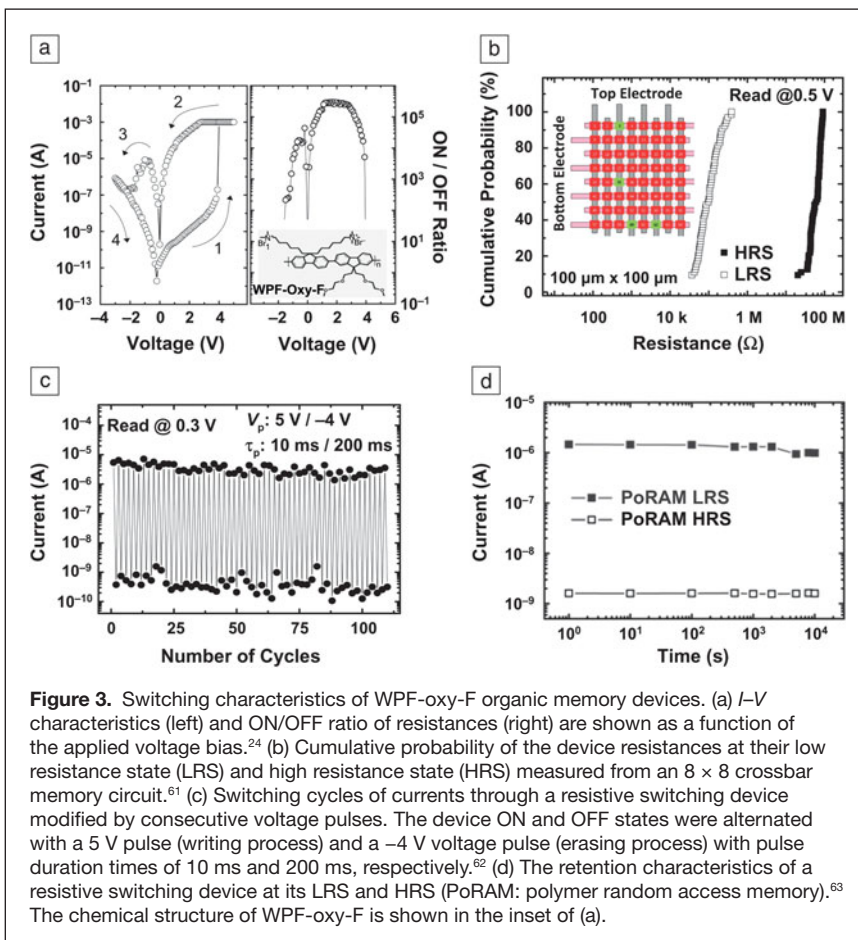


Figure 3. Switching characteristics of WPF-oxy-F organic memory devices. (a) I - V characteristics (left) and ON/OFF ratio of resistances (right) are shown as a function of the applied voltage bias.²⁴ (b) Cumulative probability of the device resistances at their low resistance state (LRS) and high resistance state (HRS) measured from an 8×8 crossbar memory circuit.⁶¹ (c) Switching cycles of currents through a resistive switching device modified by consecutive voltage pulses. The device ON and OFF states were alternated with a 5 V pulse (writing process) and a -4 V voltage pulse (erasing process) with pulse duration times of 10 ms and 200 ms, respectively.⁶² (d) The retention characteristics of a resistive switching device at its LRS and HRS (PoRAM: polymer random access memory).⁶³ The chemical structure of WPF-oxy-F is shown in the inset of (a).

The performance of organic memory devices is also evaluated in terms of sweep endurance, write-read-erase-read cycle tests, switching cycles with consecutive pulses, and retention tests. Figure 3c shows switching cycle data of a memory device measured with consecutive voltage pulses.⁶² ON and OFF states were alternated with a 5 V pulse (writing) and a -4 V voltage pulse (erasing) with pulse duration times of 10 ms and 200 ms, respectively.⁶² The typical time to write and erase conventional flash memory is several milliseconds. It is essential to develop fast organic memory systems that are rewritable in times of a microsecond or less. In this regard, some organic resistive memory devices have been demonstrated with a voltage pulse of tens of nanoseconds.⁴

Figure 3d shows retention characteristics, where the HRS and LRS values were sustained for 10^4 s without any significant degradation.⁶³ The memory retention is still not comparable to commercial nonvolatile memory devices, and significant research efforts are being devoted to improvements. Moreover, the device endurance cycles are typically limited to several hundreds to several thousands of cycles, which also limits their practical applications. This problem may arise from gradual electrical degradation by oxygen or moisture in air, and thus additional passivation or packaging processes may be a solution for this issue. The fabrication cost of organic memory is considerably cheaper than flash or other conventional silicon-based electronics because cost-effective fabrication processes such as spin-coating or printing are available for organic electronics.

Integrated circuit architectures

The crossbar structure is considered an ideal architecture for the integration of organic memory circuits (Figure 4a).⁶¹ All the overlapping regions between bottom (word lines) and top electrodes (bit lines) are referred to as information storage cells in this structure. However, a simple crossbar array in a passive matrix (called 1R array), including only information storage cells, has cross-talk interference that occurs because of the leakage of current paths (called sneak paths) through neighboring cells with low resistances^{39,41} or an excess of current that may induce electrical damage.⁶⁴ The solution to prevent cross-talk is to introduce a switching component (transistor or diode) to each cell.^{16,38,63,65} The International Technology Roadmap for Semiconductors (ITRS) also emphasized the combination of a diode or transistor with a resistor in a single chip.⁶⁶ The architecture of one diode and one resistor (1D1R)^{38,40,67-70} or one transistor and one resistor (1T1R)^{63,64} can also improve reading accessibility in an integrated memory array structure. A single element that simultaneously exhibits both switching and diode behavior would be simpler than those of

the 1D1R structure with two elements in a unit cell.^{16,32,71} A phase-separated blend of organic ferroelectric and semiconducting polymers has shown resistive switching behavior with rectifying properties.^{16,71} The improved reading capability will enable the production of high-density cross-talk-free organic memory devices integrated into an array architecture.¹⁶

Vertically stacked three-dimensional (3D) architectures greatly increase memory cell density.^{55,72-76} Figure 4b shows a 3D stacked organic memory device in the crossbar architecture, using a composite of polyimide (PI) and 6-phenyl-C61 butyric acid methyl ester (PCBM).⁷³ A spin-coating process was used to stack the active polymer layers. Once the composite layer was cured at a high temperature, subsequent spin-coating was carried out because of the thermal and chemical robustness of the PI film. The transmission electron microscopy image in Figure 4b shows the well-separated stacking of each layer. In this study, an overall device yield of 83.3% was observed for memory cells (160 well-operating cells in 192 fabricated cells in three 8×8 crossbar layers).⁷³

The use of conventional microlithography techniques is not trivial for organic memory devices because they usually include a solution process for developing or removing the photoresist, which may dissolve the active organic layer previously coated onto the substrate. To overcome this problem and downscale cell size, a direct metal transfer (DMT) method has been introduced for patterning organic memory devices.^{61,77} DMT is a

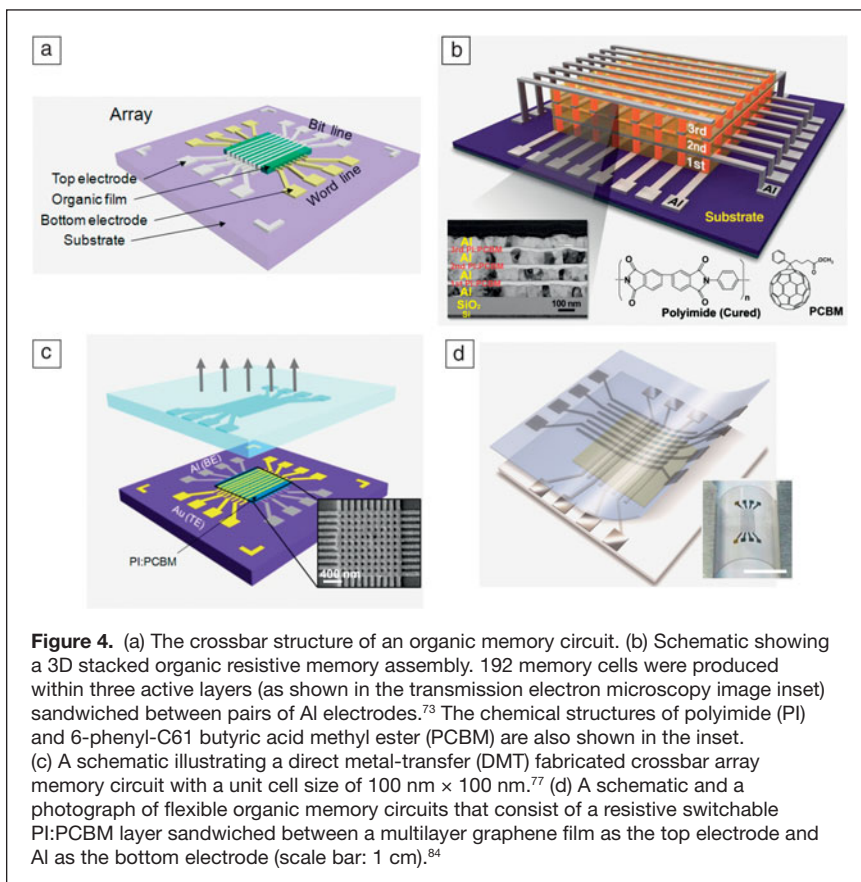


Figure 4. (a) The crossbar structure of an organic memory circuit. (b) Schematic showing a 3D stacked organic resistive memory assembly. 192 memory cells were produced within three active layers (as shown in the transmission electron microscopy image inset) sandwiched between pairs of Al electrodes.⁷³ The chemical structures of polyimide (PI) and 6-phenyl-C61 butyric acid methyl ester (PCBM) are also shown in the inset. (c) A schematic illustrating a direct metal-transfer (DMT) fabricated crossbar array memory circuit with a unit cell size of $100 \text{ nm} \times 100 \text{ nm}$.⁷⁷ (d) A schematic and a photograph of flexible organic memory circuits that consist of a resistive switchable PI:PCBM layer sandwiched between a multilayer graphene film as the top electrode and Al as the bottom electrode (scale bar: 1 cm).⁸⁴

nonaqueous process used to transfer metal patterns from a stamp directly onto a polymer layer at a low temperature and pressure (100°C and 2 MPa, respectively). This method can minimize the damage to the organic layer, ultimately enabling 3D stackable device applications. Figure 4c shows a schematic of DMT fabricated crossbar array memory devices. The inset image is the scanning electron microscopy image of fabricated PI:PCBM organic memory devices with a cell size of 100 nm.⁷⁷

Flexible electronic circuits have important potential applications, such as foldable and stretchable electronics. A variety of technologies have been conceived and applied to achieve flexibility in organic and inorganic electronics that require information storage components.^{35,72,78–82} Indeed, the use of flexible memories based on organic materials is important because of their simplicity, low manufacturing costs, and flexibility. Several studies on organic memory devices fabricated on flexible substrates have been conducted.^{35,78,82,83} Figure 4d shows a schematic of 8 × 8 array flexible organic memory devices, which consist of a PI:PCBM layer sandwiched between a multi-layer graphene film as the top electrode and Al as the bottom electrode.⁸⁴ The photograph in Figure 4d shows a semi-transparent and flexible organic memory device.

Nowadays, printable organic materials and devices are one of the important emerging technologies for the mass production of large flexible electronics. For instance, a printed crossbar Cu/CuPc/PEDOT:PSS/Ag system was recently demonstrated on a flexible substrate, and bi-stable switching behavior was observed in the device.⁸⁵ Printability is necessary for item-level radio-frequency identification (RFID) tags, large-area sensors, flexible displays, and many other emerging electronic applications because it allows simpler, cheaper, and faster fabrication of electronics compared with the fabrication of conventional semiconductor devices. Conductive polymer films^{86,87} as an alternative to metal films will further improve the bending stability of flexible electronics.

Summary and outlook

Organic resistive memory materials have attracted substantial attention due to their merits, such as nonvolatile memory characteristics, low fabrication cost, scalability, and flexibility. However, despite considerable research progress, there are still significant challenges for practical applications. For instance, it is still difficult to fully understand the resistance switching mechanisms. The organic memory devices demonstrated so far require great improvement to optimize switching characteristics such as retention time, cycling endurance, and switching speed. By resolving the scientific and technical issues described previously, organic memory technology will facilitate the development of new flexible memory applications.

Acknowledgments

The authors thank T.L.'s group members, including Byungjin Cho, Sunghoon Song, and Yongsung Ji, who are working on this topic. T.L. also acknowledges partial support from the Korean National Research Laboratory program and National

Core Research Center grant from the Korean Ministry of Education, Science, and Technology. Y.C. acknowledges the support of the Defense Advanced Research Projects Agency (DARPA) under the program "Physical Intelligence," and by the Air Force Office of Scientific Research (AFOSR) under the program "Bio-inspired intelligent sensing materials for *Fly-by-Feel* autonomous vehicle."

References

1. Y. Yang, J. Ouyang, L. Ma, R.J.H. Tseng, C.W. Chu, *Adv. Funct. Mat.* **16**, 1001 (2006).
2. J.C. Scott, L.D. Bozano, *Adv. Mater.* **19**, 1452 (2007).
3. Q.-D. Ling, D.-J. Liaw, C. Zhu, D. S.-H. Chan, E.-T. Kang, K.-G. Neoh, *Prog. Polym. Sci.* **33**, 917 (2008).
4. J. Ouyang, C.-W. Chu, C.R. Szmanda, L. Ma, Y. Yang, *Nat. Mater.* **3**, 918 (2004).
5. B. Mukherjee, A.J. Pal, *Org. Electron.* **7**, 249 (2006).
6. J. Lin, D. Ma, *Org. Electron.* **10**, 275 (2009).
7. C.W. Chu, J. Ouyang, J.H. Tseng, Y. Yang, *Adv. Mater.* **17**, 1440 (2005).
8. P.Y. Lai, J.S. Chen, *Appl. Phys. Lett.* **93**, 153305 (2008).
9. B.-O. Cho, T. Yasue, H. Yoon, M.-S. Lee, I.-S. Yeo, U.I. Chung, J.-T. Moon, B.-I. Ryu, *IEEE Int. Electron Devices Meeting* (2006); doi:10.1109/IEDM.2006.346729.
10. G. Liu, Q.-D. Ling, E.-T. Kang, K.-G. Neoh, D.-J. Liaw, F.-C. Chang, C.-X. Zhu, D.S.-H. Chan, *J. Appl. Phys.* **102**, 024502 (2007).
11. G. Liu, Q.-D. Ling, E.Y.H. Teo, C.-X. Zhu, D.S.-H. Chan, K.-G. Neoh, E.-T. Kang, *ACS Nano* **3**, 1929 (2009).
12. L.D. Bozano, B.W. Kean, M. Beinhoff, K.R. Carter, P.M. Rice, J.C. Scott, *Adv. Funct. Mat.* **15**, 1933 (2005).
13. T.-W. Kim, S.-H. Oh, H. Choi, G. Wang, H. Hwang, D.-Y. Kim, T. Lee, *Appl. Phys. Lett.* **92**, 253308 (2008).
14. H.-T. Lin, Z. Pei, Y.-J. Chan, *IEEE Electron Device Lett.* **28**, 569 (2007).
15. J.-R. Chen, H.-T. Lin, G.-W. Hwang, Y.-J. Chan, P.-W. Li, *Nanotechnology* **20**, 255706 (2009).
16. K. Asadi, D.M. de Leeuw, B. de Boer, P.W.M. Blom, *Nat. Mater.* **7**, 547 (2008).
17. Z.J. Donhauser, B.A. Mantooth, K.F. Kelly, L.A. Bumm, J.D. Monnell, J.J. Stapleton, D.W. Price, A.M. Rawlett, D.L. Allara, J.M. Tour, P.S. Weiss, *Science* **292**, 2303 (2001).
18. G. Peng, D. Yuan-Wei, J. Xin, L. Yin-Xiang, X. Wei, *IEEE Electron Device Lett.* **28**, 572 (2007).
19. C.N. Lau, D.R. Stewart, R.S. Williams, M. Bockrath, *Nano Lett.* **4**, 569 (2004).
20. B.C. Das, A.J. Pal, *Org. Electron.* **9**, 39 (2008).
21. H. Carchano, R. Lacoste, Y. Segui, *Appl. Phys. Lett.* **19**, 414 (1971).
22. H.K. Henisch, W.R. Smith, *Appl. Phys. Lett.* **24**, 589 (1974).
23. S.L. Lim, Q. Ling, E.Y.H. Teo, C.X. Zhu, D.S.H. Chan, E.T. Kang, K.G. Neoh, *Chem. Mater.* **19**, 5148 (2007).
24. T.-W. Kim, S.-H. Oh, H. Choi, G. Wang, H. Hwang, D.-Y. Kim, T. Lee, *IEEE Electron Device Lett.* **29**, 852 (2008).
25. H.S. Majumdar, A. Bandyopadhyay, A. Bolognesi, A.J. Pal, *J. Appl. Phys.* **91**, 2433 (2002).
26. Y. Sadaoka, Y. Sakai, *J. Chem. Soc., Faraday Trans. 2* **72**, 1911 (1976).
27. S.H. Kim, K.S. Yook, J. Jang, J.Y. Lee, *Synth. Met.* **158**, 861 (2008).
28. A. Laiho, H.S. Majumdar, J.K. Baral, F. Jansson, R. Osterbacka, O. Ikkala, *Appl. Phys. Lett.* **93**, 203309 (2008).
29. F. Li, T.W. Kim, W. Dong, Y.-H. Kim, *Appl. Phys. Lett.* **92**, 011906 (2008).
30. F. Li, D.-I. Son, S.-M. Seo, H.-M. Cha, H.-J. Kim, B.-J. Kim, J.H. Jung, T.W. Kim, *Appl. Phys. Lett.* **91**, 122111 (2007).
31. D.T. Simon, M.S. Griffo, R.A. DiPietro, S.A. Swanson, S.A. Carter, *Appl. Phys. Lett.* **89**, 133510 (2006).
32. F. Verbakel, S.C.J. Meskers, R.A.J. Janssen, *Chem. Mater.* **18**, 2707 (2006).
33. Y. Song, Q.D. Ling, S.L. Lim, E.Y.H. Teo, Y.P. Tan, L. Li, E.T. Kang, D.S.H. Chan, C. Zhu, *IEEE Electron Device Lett.* **28**, 107 (2007).
34. S. Paul, A. Kanwal, M. Chhowalla, *Nanotechnology* **17**, 145 (2006).
35. D.-I. Son, J.-H. Kim, D.-H. Park, W.K. Choi, F. Li, J.H. Ham, T.W. Kim, *Nanotechnology* **19**, 055204 (2008).
36. A. Bandyopadhyay, A.J. Pal, *Appl. Phys. Lett.* **84**, 999 (2004).
37. B. Cho, T.-W. Kim, M. Choe, G. Wang, S. Song, T. Lee, *Org. Electron.* **10**, 473 (2009).
38. B. Cho, T.-W. Kim, S. Song, Y. Ji, M. Jo, H. Hwang, G.-Y. Jung, T. Lee, *Adv. Mater.* **22**, 1228 (2010).
39. M.J. Lee, Y. Park, D.S. Suh, E.H. Lee, S. Seo, D.C. Kim, R. Jung, B.S. Kang, S.E. Ahn, C.B. Lee, D.H. Seo, Y.K. Cha, I.K. Yoo, J.S. Kim, B.H. Park, *Adv. Mater.* **19**, 3919 (2007).
40. H.-T. Lin, Z. Pei, J.-R. Chen, Y.-J. Chan, *IEEE Electron Device Lett.* **30**, 18 (2009).
41. I.G. Baek, D.C. Kim, M.J. Lee, H.J. Kim, E.K. Yim, M.S. Lee, J.E. Lee, S.E. Ahn, S. Seo, J.H. Lee, J.C. Park, Y.K. Cha, S.O. Park, H.S. Kim, I.K. Yoo, U.I. Chung,

- J.T. Moon, B.I. Ryu, *IEEE Int. Electron Devices Meeting* (2005), doi:10.1109/IEDM.2005.1609462.
42. H.S. Nalwa, *Ferroelectric Polymers: Chemistry, Physics, and Applications* (Marcel Dekker, New York, 1995).
43. G. Dearnaley, D.V. Morgan, A.M. Stoneham, *J. Non-Cryst. Solids* **4**, 593 (1970).
44. G. Dearnaley, A.M. Stoneham, D.V. Morgan, *Rep. Prog. Phys.* **33**, 1129 (1970).
45. L.F. Pender, R.J. Fleming, *J. Appl. Phys.* **46**, 3426 (1975).
46. Y. Segui, B. Ai, H. Carchano, *J. Appl. Phys.* **47**, 140 (1976).
47. W. Hwang, K.C. Kao, *J. Chem. Phys.* **60**, 3845 (1974).
48. W.-J. Joo, T.-L. Choi, K.-H. Lee, Y. Chung, *J. Phys. Chem. B* **111**, 7756 (2007).
49. S. Sivaramakrishnan, P.-J. Chia, Y.-C. Yeo, L.-L. Chua, P.K.H. Ho, *Nat. Mater.* **6**, 149 (2007).
50. A. Carbone, B.K. Kotowska, D. Kotowski, *Phys. Rev. Lett.* **95**, 236601 (2005).
51. P. Mark, W. Helfrich, *J. Appl. Phys.* **33**, 205 (1962).
52. S. Das, A.J. Pal, *Appl. Phys. Lett.* **76**, 1770 (2000).
53. J.G. Simmons, R.R. Verderber, *Proc. R. Soc. London, Ser. A* **301**, 77 (1967).
54. V.S. Reddy, S. Karak, A. Dhar, *Appl. Phys. Lett.* **94**, 173304 (2009).
55. J.-G. Park, W.-S. Nam, S.-H. Seo, Y.-G. Kim, Y.-H. Oh, G.-S. Lee, U.-G. Paik, *Nano Lett.* **9**, 1713 (2009).
56. R.S. Potember, T.O. Poehler, D.O. Cowan, *Appl. Phys. Lett.* **34**, 405 (1979).
57. E.I. Kamitsos, C.H. Tzimis, W.M. Risen, *Solid State Commun.* **42**, 561 (1982).
58. A.J. Kronemeijer, H.B. Akkerman, T. Kudernac, B.J.V. Wees, B.L. Feringa, P.W.M. Blom, B.D. Boer, *Adv. Mater.* **20**, 1467 (2008).
59. E.Y.H. Teo, Q.D. Ling, Y. Song, Y.P. Tan, W. Wang, E.T. Kang, D.S.H. Chan, C. Zhu, *Org. Electron.* **7**, 173 (2006).
60. J.H.A. Smits, S.C.J. Meskers, R.A.J. Janssen, A.W. Marsman, D.M. de Leeuw, *Adv. Mater.* **17**, 1169 (2005).
61. T.-W. Kim, K. Lee, S.-H. Oh, G. Wang, D.-Y. Kim, G.-Y. Jung, T. Lee, *Nanotechnology* **19**, 405201 (2008).
62. T.-W. Kim, H. Choi, S.-H. Oh, M. Jo, G. Wang, B. Cho, D.-Y. Kim, H. Hwang, T. Lee, *Nanotechnology* **20**, 025201 (2009).
63. T.-W. Kim, H. Choi, S.-H. Oh, G. Wang, D.-Y. Kim, H. Hwang, T. Lee, *Adv. Mater.* **21**, 2497 (2009).
64. K. Kinoshita, K. Tsunoda, Y. Sato, H. Noshiro, S. Yagaki, M. Aoki, Y. Sugiyama, *Appl. Phys. Lett.* **93**, 033506 (2008).
65. J.C. Scott, *Science* **304**, 62 (2004).
66. International Technology Roadmap For Semiconductors (2007). *Emerging research devices*. (Semiconductor Industry Association, International Sematech, Austin, TX 2007).
67. S. Möller, C. Perlov, W. Jackson, C. Taussig, S.R. Forrest, *Nature* **426**, 166 (2003).
68. E.Y.H. Teo, C. Zhang, S.L. Lim, E.-T. Kang, D.S.H. Chan, C. Zhu, *IEEE Electron Device Lett.* **30**, 487 (2009).
69. S.-E. Ahn, B.S. Kang, K.H. Kim, M.-J. Lee, C.B. Lee, S.G., C.J. Kim, Y. Park, *IEEE Electron Device Lett.* **30**, 550 (2009).
70. M.-J. Lee, S.I. Kim, C.B. Lee, H. Yin, S.-E. Ahn, B.S. Kang, K.H. Kim, J.C. Park, C.J. Kim, I. Song, S.W. Kim, G. Stefanovich, J.H. Lee, S.J. Chung, Y.H. Kim, Y. Park, *Adv. Funct. Mater.* **19**, 1587 (2009).
71. K. Asadi, M. Li, N. Stingelin, P.W.M. Blom, D.M. de Leeuw, *Appl. Phys. Lett.* **97**, 193308 (2010).
72. H.Y. Jeong, Y.I. Kim, J.Y. Lee, S.-Y. Choi, *Nanotechnology* **21**, 115203 (2010).
73. S. Song, B. Cho, T.-W. Kim, Y. Ji, M. Jo, G. Wang, M. Choe, Y.H. Kahng, H. Hwang, T. Lee, *Adv. Mater.* **22**, 5048 (2010).
74. W.L. Kwan, R.J. Tseng, W. Wu, Q. Pei, Y. Yang, *IEEE Int. Electron Devices Meeting* (2007); doi:10.1109/IEDM.2007.4418911.
75. W.L. Kwan, R.J. Tseng, Y. Yang, *Philos. Trans. R. Soc., A* **367**, 4159 (2009).
76. C. Kügeler, M. Meier, R. Rosezin, S. Gilles, R. Waser, *Solid-State Electron.* **53**, 1287 (2009).
77. J.J. Kim, B. Cho, K.S. Kim, T. Lee, G.Y. Jung, *Adv. Mater.* **23**, 2104 (2011).
78. L. Li, Q.-D. Ling, S.-L. Lim, Y.-P. Tan, C. Zhu, D.S.H. Chan, E.-T. Kang, K.-G. Neoh, *Org. Electron.* **8**, 401 (2007).
79. S. Kim, Y.-K. Choi, *Appl. Phys. Lett.* **92**, 223508 (2008).
80. S. Lee, H. Kim, D.-J. Yun, S.-W. Rhee, K. Yong, *Appl. Phys. Lett.* **95**, 262113 (2009).
81. H.Y. Jeong, J.Y. Kim, J.W. Kim, J.O. Hwang, J.-E. Kim, J.Y. Lee, T.H. Yoon, B.J. Cho, S.O. Kim, R.S. Ruoff, S.-Y. Choi, *Nano Lett.* **10**, 4381 (2010).
82. Y. Ji, B. Cho, S. Song, T.-W. Kim, M. Choe, Y.H. Kahng, T. Lee, *Adv. Mater.* **22**, 3071 (2010).
83. T. Sekitani, T. Yokota, U. Zschieschang, H. Klauk, S. Bauer, K. Takeuchi, M. Takamiya, T. Sakurai, T. Someya, *Science* **326**, 1516 (2009).
84. Y. Ji, S. Lee, B. Cho, S. Song, T. Lee, *ACS Nano* **5**, 5995 (2011).
85. K. Jian, R. Li, H. Wang, J. Zhang, D. Gamota, *Mater. Sci. Eng., B* **167**, 12 (2010).
86. A.J. Heeger, *Angew. Chem. Int. Ed.* **40**, 2591 (2001).
87. A.G. MacDiarmid, *Angew. Chem. Int. Ed.* **40**, 2581 (2001).
88. R. Waser, M. Aono, *Nat. Mater.* **6**, 833 (2007).
89. Q.D. Ling, S.L. Lim, Y. Song, C.X. Zhu, D.S.H. Chan, E.T. Kang, K.G. Neoh, *Langmuir* **23**, 312 (2007).
90. L.-H. Xie, Q.-D. Ling, X.-Y. Hou, W. Huang, *J. Am. Chem. Soc.* **130**, 2120 (2008). □



2012 New Diamond and
Nano Carbons Conference

May 20-24, 2012 Conrad San Juan Condado Plaza – San Juan, Puerto Rico

REGISTER BY MAY 3, 2012 AND SAVE!

Preregistration opens mid-February



SAVE THE DATE

Join us for the fourth international New Diamond and Nano Carbons Conference (NDNC 2012). Hosted in beautiful San Juan, Puerto Rico, the conference will present high-impact scientific and technological advances, along with critical developments to enable the application of diamond, carbon nanostructures and related materials in a diverse range of products. This year's conference will feature plenary sessions, parallel topical sessions, poster sessions and an industrial exhibition. Mark your calendars today and plan to attend NDNC 2012!

SCIENTIFIC PROGRAM

The five-day conference will feature oral and poster presentations covering:

- Science and Technology of Diamond
- Synthesis and Characterization of Graphenes
- Synthesis and Characterization of Nano Carbons
- C-based Coatings and Tribological Applications
- Interfaces and Heterojunctions of C-based Materials with Other Materials
- Electronic, Optical and Acoustic Applications of C-based Materials
- Electrochemical Applications of C-based Materials
- Sensor Applications of C-based Materials
- Biological Applications of C-based Materials
- Spintronics of C-based Materials
- Business Opportunities and Commercialization

www.mrs.org/ndnc-2012